

## The Mechanism Research of Electric Current Pulse and Electromagnetic for Grain Refinement of 6181 Aluminum Alloys During Two-roll Casting

Keming Sun<sup>1</sup>, Yichong Zhang<sup>1</sup>, Yenan Lv<sup>1</sup>, Guangming Xu<sup>1\*</sup>, Xin Su<sup>1</sup>, Guoming Chi<sup>2</sup> and Chenghao Wang<sup>3</sup>

<sup>1</sup>Key Laboratory of Electromagnetic Processing of Materials, Northeastern University, Shenyang 110004, PR China

<sup>2</sup>Ruyuan Dong Guang Fine Foil Co., Ltd, Guangdong, China

<sup>3</sup>State Grid Smart Grid Research Institute, Beijing 100192, China

### Research Article

Received date: 22/04/2015

Accepted date: 04/01/2016

Published date: 08/01/2016

#### \*For Correspondence

Guangming Xu, The Key Laboratory of Electromagnetic Processing of Material, Northeastern University, Liaoning, China.

Email: xu\_gm@epm.neu.edu.cn

**Keywords:** Aluminum alloys, Two-roll casting, Force of electromagnetic, Grain sizes, Mechanical property.

#### ABSTRACT

6181 aluminum alloys sheets were produced by two-roll casting with electric current pulse and magnetic fields, and the microstructures were observed. The experimental results show that application of composite fields (ECP and EM field) during TRC process refines the microstructures of sheets more evenly and effectively by comparison with individual application of ECP or EM field. For visually explaining the phenomenon, the distribution and amplitude of the magnetic induction intensity  $B$  and electromagnetic force  $F$  were obtained by numerically simulated. In addition, an obvious improvement of tensile strength by physical fields was measured. But the maximum elongation was from action of magnetic field and reached 10.9%.

### INTRODUCTION

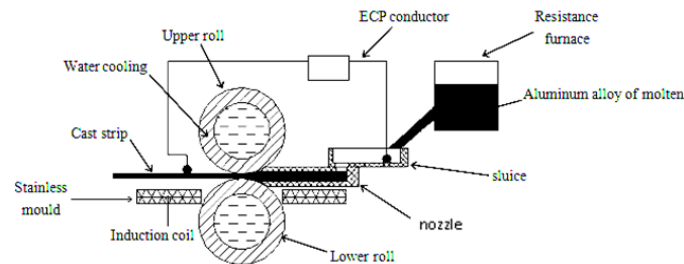
Recently, the topic of saving energy and environment protection has been paid more and more attention. In the field of materials, in the interest of reaching the above requirements, the research of light alloy and new methods of production has been a consequent trend. 6XXX aluminum alloys, which have been known as a kind of automotive materials, due to their low density, high specific strength, excellent plasticity, and good electrical conductivity [1-3]. However, how to reduce the manufacture costs and the improve formability of aluminum alloys sheets is required. In this research, the 6181 aluminum alloys sheets are produced by the two-roll casting. The twin-roll casting (TRC) is manufacturing process that makes crystallization, solidification, deformation combine into a whole, and it has been confirmed as a simple economical and effective method for production of aluminum [4,5]. In 2000, during twin roll casting of aluminum alloys, various macroscopic defects such as surface bleeds and deformation segregates were observed by Yun [6]. In 2005 and 2006, Gras [7] and Forbord [8] investigated also respectively twin roll casting of aluminum alloys and found the micro defects formation. With the development of computer, combination of simulation and test become an effective method for TRC. Chen Shou-dong [9] built the analytical model of twin-roll continuous casting aluminum thin strip solidification and provided some theoretical direction. Lee Ys [10] researched the influence of different processing parameters (symmetric nozzle and symmetric roll speed) and roll separation force for the temperature and liquid fraction, and found a good quality of the strip was experimentally obtained at a roll separation force of 6-8t. From here we see that the TRC process has attracted a batch researchers' interest.

However, how to decrease the defection and columnar crystals, enhance the mechanical property has not been well solved. In

the process of cast ingot, in order to reduce the inhomogeneity of structure, refine grain sizes and improve the mechanical property, electromagnetic fields (EM) and electric current pulse [ECP] are applied to the solidifications of organization. Cui Jian-zhong<sup>[11]</sup> suggested that the magnetic fields can refine the microstructures and improves the surface quality. Ma Jianhong<sup>[12]</sup> offered some experimental evidence that the solidification structure can be refined and the proportion of equated area can be raised by the electric current pulse. So it is a good idea to explore the effect of the EM fields and ECP for the structures in TRC process.

## MATERIALS AND METHODS

The 6181 aluminum alloys are composed by 97.1% Al, 1.2% Si, 1.0% Mg, 0.5% Fe and 0.2% Mn. The experimental procedure of TRC process without and with EPC field and EM field as shown as **Figure 1**. First of all, the 99.85wt% pure Al ingot and Al-Si alloys (Silicon accounted for the proportion of 20% in alloys) were let into resistance furnace that manually set up to 750°C.



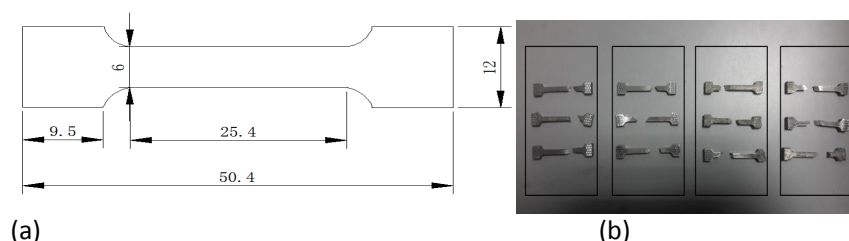
**Figure 1.** Schematic diagram of the TRC process with ECP and EM fields.

About 2 hours later, the Al and Al-Si alloys were melted into liquid. And then the furnace temperature was reduced to 720°C; the 99.95% pure Mg blocks were added to the homogeneous melt and stirred until complete disappearing. And the furnace temperature was reduced once more to 690°C (It is the pouring temperature). 30 min later, the solution was slagged off and degassed. At the beginning of TRC experiments, the molten aluminum alloy was poured into the sluice and nozzle which were pre-heated to 400°C. At the same time, cooling water was passed through the twin rolls to reduce melt temperature, and the ECP and EM fields were added to the TRC process. The ECP parameters were optimized as follows: 20 Hz frequency and 400 I peak current. A static magnetic field with a nominal value of 0.02-0.25T was applying at cast-rolling areas. The velocity of experimental rolling is 0.7m min<sup>-1</sup>. The diameter of experimental rolling is 500mm.

Four sheets were manufactured under different forming conditions. The first sheet was formed by non-field TRC process, the second under ECP condition. The third sheet was manufactured under EM field's condition. The last one was produced under combination of ECP and static magnetic field condition.

Rectangular specimens of 20mm×10mm×5mm were cut out from the strips along the rolling direction. The surfaces of specimens were mechanical polished and coated film by 3.4ml HBF<sub>4</sub> and 96.6ml H<sub>2</sub>O. The optical microscope (OM) was used to observe grain organization of specimen.

Multiple sample tensile testing was performed using a SHIMADZU AG-X100KN Material Testing Machine along the rolled direction at room temperature under constant deformation speed of 0.5 mm min<sup>-1</sup>. The dimension for the tensile specimen and specimens of fracture are shown in **Figure 2**. And the morphology of fracture was observed by a SSX-550 scanning electron microscope (SEM).

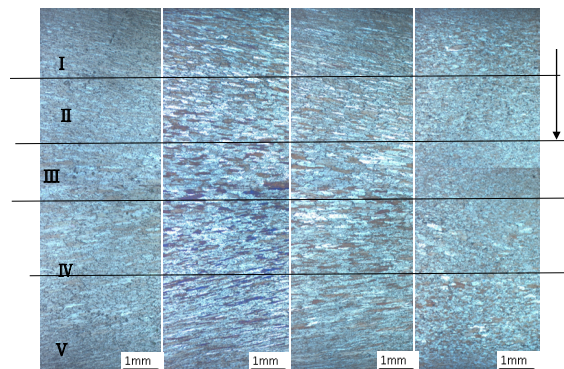


**Figure 2.** Diagrammatic sketch of pinch force

## RESULTS

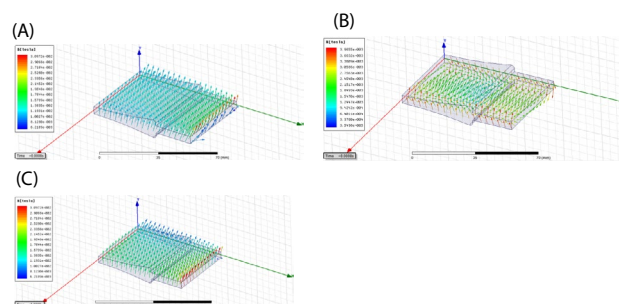
The distribution of grains for 6181 aluminum alloys sheets is shown as **Figure 1**. In order to observe the changes of grains, the whole area is divided into V parts. I and V corresponds with chilling area in the surfaces; II and IV are the columnar crystals zone; III is the equiax crystal zone at the centrality. The arrows point out the direction of solidification and pressure. The morphology of surfaces without the physical fields is typical broken grains, and they distribute along a 5° - 10° angle difference with horizontal direction. The length of slope with the ECP or EM field has a little increase. However, when the ECP and EM field are together applied to TRC process, the slope disappears nearly (**Figure 3**). At the columnar crystals zone, the physical fields make the grain distribution more uniformity. Based on the fundamentals of solidification, the dendritic/cellular crystals grow coincident

with solidification direction; but in the TRC process, the crystals are flattened by the pressure from two rolls. Moreover, under the action of physical fields, fat grains become gradually thin. And the quiax crystal zone is amplified and grain sizes are refined obviously with ECP and EM field. And effect of the composite fields is better than them alone for grain refinement.

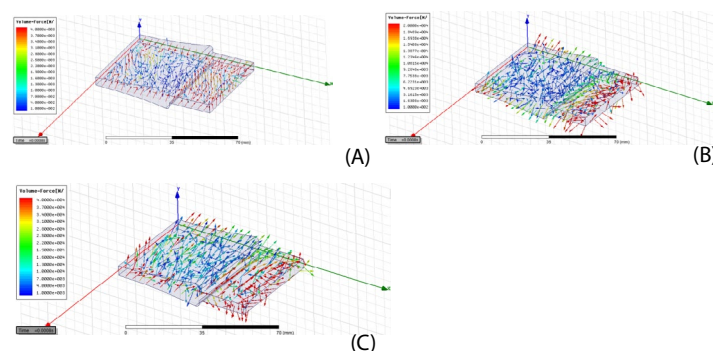


**Figure 3.** Microstructure of the strip obtained from TRC process. Micrograph (a) corresponds to the conventional TRC strip, and micrograph (b), (c) and (d) corresponds respectively to with ECP, EM and complex fields.

To visualize directly the influence of magnetic force for grains growth during TRC process, the distribution of magnetic induction intensity  $B$  and distribution of electromagnetic force  $F$  are simulated by the Ansoft Maxwell soft. **Figure 4** shows the distribution of magnetic induction intensity  $B$  from melt, mushy zone to solidity. According to right-hand screw rule, the  $B_1$  is generated by current in the magnetic fields, and their direction point out vertical up, and the numerical value is about 13.8 – 29.0 mT, which conforms to the actual measured value. The magnetic induction intensity  $B_2$  generated by ECP is along the clockwise direction, and their values decrease gradually from external to internal. Due to the lesser values from ECP, therefore, when the ECP and EM field together act on TRC process, values of the magnetic induction intensity  $B_3$  change little compared with  $B_1$  from magnetic alone. But their direction occur incline to the  $-z$  axle. **Figure 5** indicates distribution of electromagnetic force  $F_1$ , which is related with magnetic induction intensity  $B$  and current  $I$ . In the magnetic fields, the direction of electromagnetic force  $F$  is outward. The electromagnetic force  $F_2$  that is produced by ECP points out interior at the time. Here, it should be emphasize that the  $F_2$  is different with  $F_1$ , with change of times, direction of  $F_1$  is almost constant, but direction of the  $F_2$  turns towards inside and outside alternate change. This means that electromagnetic force  $F_3$  has some change in the direction when ECP and EM field meet. And the  $F_3$  is greater than  $F_1$  and  $F_2$ .

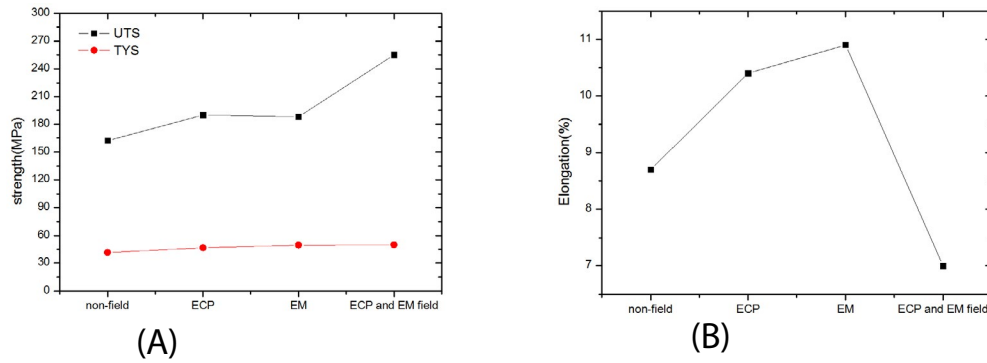


**Figure 4.** Distribution of magnetic induction intensity  $B$  from (a) EM field (b) ECP (c) complex fields



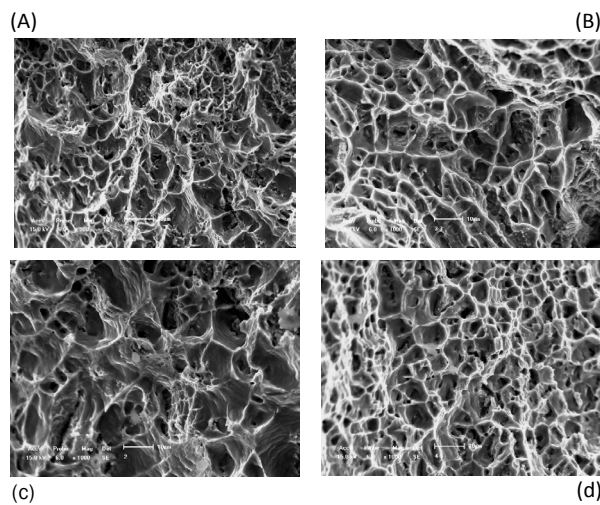
**Figure 5.** Distribution of electromagnetic force  $F$  produced by (a) EM field (b) ECP (c) complex field

In the presence and absence of the physical fields, the mechanical properties of 6181 aluminum alloys, including the ultimate tensile strength  $\sigma_b$  (UTS), yield strength  $\sigma_{0.2}$  (YS) as well as the elongation, are shown in the **Figure 6a and 6b**. As observed in the **Figure 6a** the UTS under the physical fields are superior to one of materials prepared in the normal TRC process. And the composite fields are more effective than separate them. The YS change little. However, curve of the elongation that has been obtained from cracked samples are different from the rule of UTS and YS. The maximum elongation is from with magnetic field and reaches 10.9%.



**Figure 6.** The variation trends of the elongation, UTS and TYS values of tensile specimens: (a) Elongation (b) UTS and TYS values

**Figures 7a and d** shows the fracture surface morphology of tensile sample in the longitudinal direction. All these specimens show the typical ductile fracture. The depth and sizes of **(Figure 7b and c)** dimple is greater than others. However, the fracture surface morphology of tensile sample with composite field is small and low.

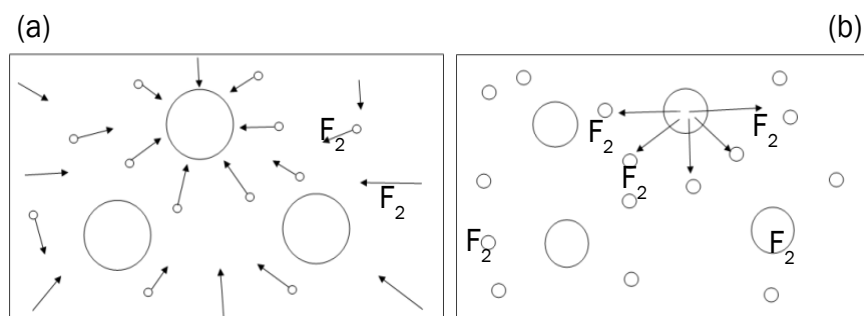


**Figure 7.** SEM micrographs of the fracture surface morphology: (a) the non-field sheets (b) ECP sheets (c)EM field sheets (d) complex field sheets

## DISCUSSION

First of all the effect and mechanism of ECP on the grain refining should be discussion. The above experimental result **(Figure 3b)** shows that the application of ECP during TRC process can modify the morphology and sizes of grains. The result should be attributed to two reasons. One of the reasons is explained in the aluminum alloy melt stage.

A great deal of atom groups of short-range order and meeting and parting changing consist in metal melt. Gather of them depend on electrostatic effect [13]. When the ECP acts on the metal melt, the pinch force  $F_2$  is produced as **Figure 5b**. Under action of the force  $F_2$ , the adjoining free atoms and small short-range order atom groups are possible to come into being great atom groups as shown as **Figure 8a**. When the great atom groups reach a particular size, the stable embryos will be taken shape, and they enhance nucleation probability [14]. When direction of the  $F_2$  points out outside, the instability atom groups are broken as free atoms as **Figure 8b**, but the rate of gather overtops the rate of dispersed. The other one reason comes from solidification stage. According to Barnak the pinch force  $F_2$  can be expressed as the following formula [15].



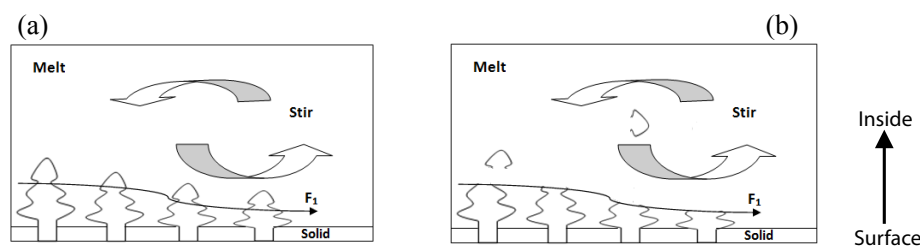
**Figure 8.** The dimension of the tensile specimen, and specimen of fracture

$$\sigma = \nu[\mu J^2(r^2 - a^2)/4] \quad (1)$$

Where  $\nu$  is Poisson's ratio,  $\mu$  the permeability,  $J$  the current density,  $r$  the specimen radius and  $a$  distance from an arbitrary point to centrality of sample. The maximum pinch force  $F_2$  can be only 1 KPa by their compute, which accord with the result of simulation. They are enough strong to stir the melt. The crystal nuclei formed in the surface of two rolls are brought to melt and become new nucleation points. Therefore, the grains are refined. Secondly, in the EM fields, the convection is the main reason to refine grains. The dendrite begins growth from surface of two rolls to the inside of melt. According to the magnetic force  $F_1$  as **Figure 5a**, their direction is perpendicular to direction of dendrite growth. When the melt is stirred by the  $F_1$ , the strong convection is produced. The shearing force is formed under the convection acting on liquid - solid interface; when their strength exceeds resistance of dendrite arms, the dendrite arms can be broken. Fragments of dendrite arms are brought into the inside of melt as **Figure 9**. A part of fragments is remelted; the other part becomes neterogeny cores of equiax crystal.

The magnetic mechanism had been proved by a large number of researchers [16-19]. Moreover, the dendrite growth and break with magnetic field are further observed and certified by experiments [20]. However, when the ECP and EM fields are together added to the TRC process, the ECP and EM fields arises reaction, and the magnetic force  $F_3$  are generated, which can be expressed as following formula

$$F=J*B$$



**Figure 9.**Diagrammatic sketch of dendrites break and melt stirring

Where  $B$  is magnetic induction intensity and  $J$  is the current density. The melt and liquid - solid interface suffer not only  $F_1$  and  $F_2$  but also  $F_3$ . Besides, the value of  $F_3$  exceeds  $F_1$  and  $F_2$ , and the shake of convection is more violent. Therefore, the rate of nucleation is increased. Grains are refined and equiaxed. As is well known, the tensile strength increases with the decrease of grain size [21]. Gao [22] indicated that the refinement of grains could suppress deformation by twinning and sliding. Obviously, for comparison these UTS and YS values, it is contrary to the sizes of grains.

## CONCLUISON

The influence of electric current pulse and magnetic field for grain refining and distribution of 6181 aluminum alloys during two-roll casting was investigated. The experimental results show that application of composite fields (ECP and EM field) during TRC process refines the microstructures of sheets more evenly and effectively by comparison with individual application of ECP or EM field. Distribution and amplitude of the magnetic induction intensity  $B$  and electromagnetic force  $F$  were numerically simulated. The results reveal that melt and liquid- solid interface suffer more force from composite fields than individual one. And the tensile strength was enhanced when physical fields were applied to TRC process. However, effect of EM field for elongation is best and the value reaches 10.9%.

## REFERENCES

1. Zhang HT and Cui JZ. Production of super-high aluminum alloy billets by low frequency electromagnetic casting, T Nonferr Metal Soc 2011; 21:2134-2139.
2. Zuo Y, et al. Effects of low frequency electromagnetic field on the as-cast microstructures and mechanical properties of super high strength aluminum alloys, Mat Sci Eng A 2005; 408: 176-181.
3. Laha T, et al. Synthesis of bulk nanostructured aluminum alloy component through vacuum plasma sprays technique, Acta Mater 2005; 53:5429-5438.
4. Watari H, et al. Warm deep drawing of wrought magnesium alloy sheets produced by semi-solid roll strip-casting process, Int J Mach Tool Manu 2006; 46:1233-1237.
5. Chen ZW, et al. Homogenization of twin-roll cast A8006 alloy, T Nonferr Metal Soc 2012; 22:1280-1285.
6. Yun M. Twin rolls casting of aluminum alloys. Mat Sci Eng A 2000; 280:116-123.
7. Gras CH, et al. Micro defects formation during the twin-roll casting of Al-Mg-Mn aluminum alloys, J Mater Process Tech 2005; 167:62-72.
8. Forbord B, et al. The formation of surface segregates during twin roll casting of aluminum alloys. Mat Sci Eng A 2006; 415:12-20.

9. Shou-dong C and Jing-chao C .Simulation of microstructures in solidification of aluminum twin-roll casting, T. Nonferr Metal Soc 2012; 22:1452-1456.
10. Lee YS, et al. Process parameters and roll separation force in horizontal twin roll casting of aluminum alloys. J Mater Process Tech 2015; 218:48-56.
11. Jian-zhong C, et al. DC casting of light alloys under magnetic fields. T Nonferr Metal Soc, 2010; 20:2046-2050.
12. Jianhong Ma, et al Grain refinement of pure Al with different electric current pulse modes, Mater Lett 2009; 63:142-144.
13. Jiangzhong W. The research of treating technology with electro-pulse modification and the hypothesis of liquid metal cluster structure, PhD Dissertation University of Science and Technology Beijing 1998.
14. Jinhui F. Fundamental Research on Refinement of solidification Structure of Metals with Pulse Current. PhD Dissertation Shanghai University 2005.
15. Barnak JP, et al. Colony (Grain) Size Reduction in Eutectic Pb-Sn Castings by Electroplusing. Scripta Materialia 1995; 32:879-884.
16. Dahle AK, et al. Modeling the Fluid-Flow-Induced Stress and Collapse in a Dendritic Network. Metall Mater Trans 1999; 30:287-294.
17. Dahle AK and Arnberg L.The Rheological Properties of Solidifying Aluminum Foundry Alloys. JOM, 1996;48: 34-37.
18. Dahle AK, LArnberg. Development of Strength in Solidification Aluminium Alloys. Acta Mater 1997; 45:547-559.
19. Dayong G. The growth of dendrite and composition segregation during solidification with composite physical field, PhD Dissertation Institute of Metal Research, China.
20. Li X, et al. Effect of a weak transverse magnetic field on solidification structure during directional solidification, Acta Mater 2014; 64:367-381.
21. Shao Z, et al .A new method of semi-continuous casting of AZ80 Mg alloy billets by a combination of electromagnetic and ultrasonic fields. Mater Des 2011; 32:4216-4224.
22. Gao DM, et al. Effect of ultrasonic power on microstructure and mechanical propertied of AZ91 alloy, Mat Sci Eng A 2009; 502:2-5.